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 $Cs_2Ga_2O(BO_3)_2$ 

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### **Table 2. Selected bond lengths (A)**



Symmetry codes: (i)  $x, y, 1 + z$ ; (ii)  $1 - x, 1 - y, 1 - z$ .

Data collection: *RCRYSTAN85* (Rigaku Corporation, 1985). **Data reduction:** *TEXSAN* **(Molecular Structure Corporation,**  1989). Program(s) **used to solve structure:** *SAPI91* (Fan, 1991). Program(s) **used to refine structure:** *TEXSAN.* **Molecular graphics:** *ORTEPII* **(Johnson,** 1976).

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**Lists of structure factors, anisotropic displacement parameters and complete geometry have been deposited with the IUCr (Reference:**  ASl146). **Copies may be obtained through The Managing Editor, International Union of Crystallography,** 5 Abbey Square, Chester CH1 2HU, **England.** 

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**tice with Cs atoms occupying large cavities in the lattice.** 

**Abstract** 

**This lattice is made from pairs of Ga atoms connected through a shared O atom along the [001] direction; these Ga dimers are interconnected through shared O vertices**  to trigonal  $BO<sub>3</sub>$  groups along the [010], [110] and [110] **directions, completing the tetrahedral coordination environment of each Ga atom.** 

**The structure of dicaesium digallium diorthoborate oxide constitutes a new type of oxide borate. It consists of a three-dimensional gallium-oxide-orthoborate lat-**

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# **Comment**

**Solid-state borates have been shown to crystallize with a wide variety of structures (Heller, 1986) as a result of the vast number of possible ways of combining the borate anions with the metal-atom polyhedra. The metal-atom polyhedra may be octagonal, tetrahedral or square planar, and the borate anions may be trigonal or tetrahedral orthoborates, pyroborates or combinations thereof, linear chains, or fused tings, thus affording vast numbers of structural types. Partly in order to study such new structural types and partly to search for new nonlinear optical materials (of which several examples of**  borates exist), we recently surveyed the  $Cs<sub>2</sub>O-Ga<sub>2</sub>O<sub>3</sub>$ **B203 phase diagram, the compounds in which have not been reported previously. Among the compounds discovered in this system is an oxide borate of a new**  structural type,  $Cs_2Ga_2O(BO_3)_{2}$ .

**A unit cell of the title compound is depicted in Fig. 1. The structure consists of a three-dimensional galliumoxide--orthoborate lattice that affords large vacancies within which Cs atoms are located. The lattice is built from pairs of Ga atoms that are interconnected through an 0(7) atom and which lie roughly along the [001] direction. 0(7) is an oxide anion since it is not bonded to either of the two crystallographically distinct B atoms. Each Ga atom completes its tetrahedral coordination by bonding to O atoms that constitute the two crystallographically distinct trigonal orthoborate groups. These orthoborate groups connect the Ga-- O Ga dimers roughly along the [010], [110] and**   $\overline{110}$  directions. Because the Ga- $\overline{0}$ -Ga bond angle **is relatively large [140.8 (4)°], large cavities are created** 

between groups of dimers; these also run roughly along the [110] direction. Cs atoms occupy these cavities.

All bond distances and angles are in accordance with expected values. The Cs coordination environments are irregular eight-coordinate polyhedra. The Cs-O bond distances range from  $2.885(7)$  to  $3.456(8)$  Å, with an average value of  $3.22(18)$  Å. These values are reasonable given the expected value of 3.12A obtained from the crystal radii of the eight-coordinate  $Cs^+$  ion and the four-coordinate  $O^{2-}$  ion (Shannon, 1976). Average Ga-O and B-O bond distances are 1.83(2) and  $1.37(1)$  Å, respectively, which compare well with the calculated values of 1.85 and 1.39 A. The shortest Ga- $-$ O bonds are to the oxide O(7) [1.794 (7) and  $1.822(7)$  Å] because this O atom has no electron density removed by the  $\pi$  system of the BO<sub>3</sub> groups.

The O atoms can be three-, four- or five-coordinate.  $O(3)$  is three-coordinate,  $O(1)$ ,  $O(6)$  and  $O(7)$  are fourcoordinate, and O(2), 0(4) and 0(5) are five-coordinate.



Fig. 1. Sketch of the unit cell of  $Cs_2Ga_2O(BO_3)_2$  viewed down the  $b$  axis, where the largest circles represent  $O$  atoms, the small black circles represent B atoms, and the medium-sized darkly shaded circles represent Ga atoms. Cs atoms are shown as medium-sized lightly shaded circles without bonding interactions.

# **Experimental**

A crystal of  $Cs_2Ga_2O(BO_3)_2$  was separated from a melt of the compound  $Cs_3Ga(BO_3)_2$  in an attempt to grow crystals of the latter material. The latter material was prepared by mixing

 $CsNO<sub>3</sub>$  (Johnson Matthey, 99.9%),  $Ga(NO<sub>3</sub>)<sub>3</sub>$ xH<sub>2</sub>O (Aldrich, 99.9%) and  $B_2O_3$  (Johnson Matthey, 99%) in a 3:1:1 molar ratio, grinding the mixture under hexane, heating it to 600 K overnight to decompose the nitrates, and then sintering it at 775 K for 2 d. Crystals of the title material were grown from a melt of this material, which obviated the need to grow them from the pure phase. The material was melted at 1025 K, slowly cooled at  $6K h^{-1}$  to 575 K, and then air-quenched.

reflections<br> $\theta = 15-18^{\circ}$ 

 $R_{\text{int}} = 0.051$  $\theta_{\text{max}} = 30^{\circ}$  $h=0\rightarrow 12$  $k=0\rightarrow 11$  $l = -17 \rightarrow 17$ 3 standard reflections monitored every 200 reflections intensity decay: 1.9%

#### *Crystal data*

 $\text{Cs}_2\text{Ga}_2\text{O}(\text{BO}_3)_2$  Mo  $\text{K}\alpha$  radiation<br>  $M_r = 538.87$   $\lambda = 0.71069 \text{ Å}$  $M_r = 538.87$   $\lambda = 0.71069~\text{\AA}$ <br>Monoclinic Cell parameters Monoclinic Cell parameters from 20<br> $P2_1/c$  reflections  $a = 8.889 (1)$   $\AA$   $\theta = 15-18^{\circ}$ <br>  $b = 7.998 (1)$   $\AA$   $\mu = 14.795$  mm<sup>-1</sup>  $b = 7.998(1)$  Å  $\mu = 14.795$ <br>  $c = 12.237(1)$  Å  $T = 295$  K  $c = 12.237(1)$  Å  $T = 2$ <br>  $\beta = 103.82(1)$ ° Plate  $\beta = 103.82$  (1)<sup>o</sup>  $V = 844.8(2)~\text{\AA}^3$  0.45 x 0.15 x 0.06 mm  $Z = 4$  Colorless  $D_r = 4.236$  Mg m<sup>-3</sup>

*Data collection*  Rigaku AFC-6R diffractometer  $2\theta-\omega$  scans Absorption correction: refined from  $\Delta F$ *(DIFABS;* Walker & Stuart, 1983) 2790 measured reflections 2470 independent reflections 2002 observed reflections  $[F_o^2 > 3\sigma(F_o^2)]$ 

### *Refinement*



# Table 1. *Fractional atomic coordinates and equivalent isotropic displacement parameters*  $(A^2)$

$$
U_{\mathbf{eq}} = (1/3)\Sigma_i \Sigma_j U_{ij} a_i^* a_j^* \mathbf{a}_i \mathbf{a}_j.
$$





Table 2. Selected geometric parameters (Å. °)

Symmetry codes: (i)  $x - 1$ , y, z; (ii)  $1 - x$ ,  $-y$ ,  $-z$ ; (iii)  $-x$ ,  $y - \frac{1}{2}$ ,  $\frac{1}{2} - z$ ; (iv)  $x, \frac{1}{2} - y, z - \frac{1}{2}$ ; (v)  $-x, -y, 1 - z$ ; (vi)  $1 - x, \frac{1}{2} + y, \frac{3}{2} - z$ ; (vii)  $x, \frac{1}{2} - y, \frac{1}{2} + z$ ; (viii)  $1 - x, -y, 1 - z$ ; (ix)  $1 + x, y, z$ ; (x)  $x, y, 1 + z$ .

A clear, colorless, plate-shaped crystal was mounted on a glass fiber with epoxy for X-ray structure analysis. Data were collected with a scan speed of  $16.0^{\circ}$  min<sup>-1</sup> in  $\omega$  and a scan width of  $(1.5 + 0.3\tan\theta)$ °.

*The TEXSAN* crystallographic software package (Molecular Structure Corporation, 1989) was used to solve the crystal structure. The Laue symmetry *2/m* and the systematic absences *hOl*  $(l = 2n + 1)$  and  $0k0$   $(k = 2n + 1)$  unambiguously indicated space group  $P2<sub>1</sub>/c$ . The positions of the Cs atoms were determined by a Patterson vector map and the remaining atoms were found by examination of the difference electron density maps. After isotropic refinement, the data were averaged and corrected for absorption.

Lists of structure factors, anisotropic displacement parameters and complete geometry have been deposited with the IUCr (Reference: BR1078). Copies may be obtained through The Managing Editor, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

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# **Neodymium-Doped Srs(PO4)aF and Srs(VO4)aF**

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### **Abstract**

Neodymium-doped  $Sr<sub>5</sub>(PO<sub>4</sub>)<sub>3</sub>F$  [neodymium strontium fluoride phosphate,  $(Nd, Sr)_{5}(PO_{4})_{3}F1$  and neodymiumdoped  $Sr<sub>5</sub>(VO<sub>4</sub>)<sub>3</sub>F$  [neodymium strontium fluoride vanadate,  $(Nd, Sr)_{5}(VO_{4})_{3}F$ ] crystallize in space group *P63/m* and are isostructural with calcium fluorophosphate,  $Ca<sub>5</sub>(PO<sub>4</sub>)<sub>3</sub>F$ . There are two different Sr sites in  $Sr<sub>5</sub>(XO<sub>4</sub>)<sub>3</sub>F$ , denoted  $Sr(1)$  and  $Sr(2)$ . Using singlecrystal X-ray diffraction the two structures were refined to  $R$  factors of 2.3 and 2.2%, respectively, showing that Nd is present at both Sr sites in  $(Sr, Nd)_{5}(VO_{4})_{3}F$  but only at the Sr(2) site in  $(Sr, Nd)_5 (PO_4)_3F$ .

# **Comment**

**Calcium fluorophosphate (FAP) crystals have been investigated previously as laser hosts, and lasing characteristics** of rare-earth-doped FAP were reported as early as 1968 (Ohlmann, Steinbruegge & Mazelsky, 1968). Recent developments have reopened investigations into FAP and the whole calcium-strontium fluorophosphate system for use as laser hosts. This paper presents the detailed crystal structure refinements that followed the recent successful crystal growth of neodymium-doped  $Sr<sub>5</sub>(PO<sub>4</sub>)<sub>3</sub>F$  (SFAP) and neodymium-doped  $Sr<sub>5</sub>(VO<sub>4</sub>)<sub>3</sub>F$ (SVAP).

Both crystals were grown by the Czochralski method. SVAP and SFAP are isostructural and belong to the hexagonal space group *P63/m. The* structure contains layers of  $XO_4$  tetrahedra  $(X = P, V)$  centred at two levels along [001] and  $F^-$  ions lying along the c axis, as can be seen in Fig. 1. V—O and P—O bond lengths are given in Tables 2 and 4, respectively, and show that the VO4 tetrahedra are distorted [average/(deviation) for P- $\sim$ O and V- $\sim$ O bond distances are 1.541 Å/(0.004 Å) and 1.711  $\rm \AA/(0.01\AA)$ , respectively].

There are two independent Sr sites in the structures, the first of which,  $Sr(1)$ , contains  $40\%$  of the  $Sr^{2+}$ 

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